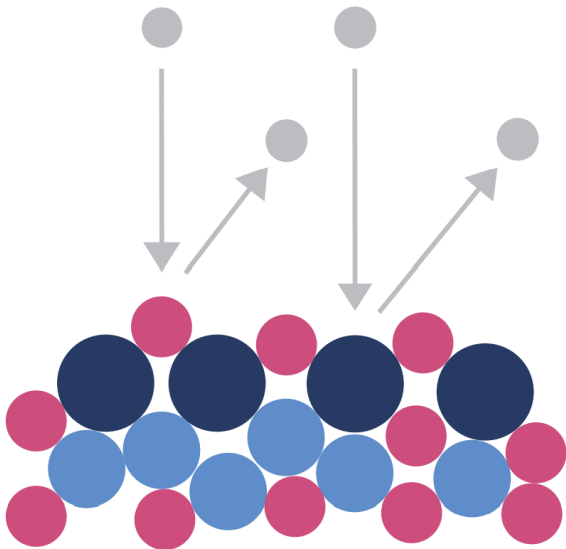


Low Energy Ion Scattering
Quantitative Surface Analysis

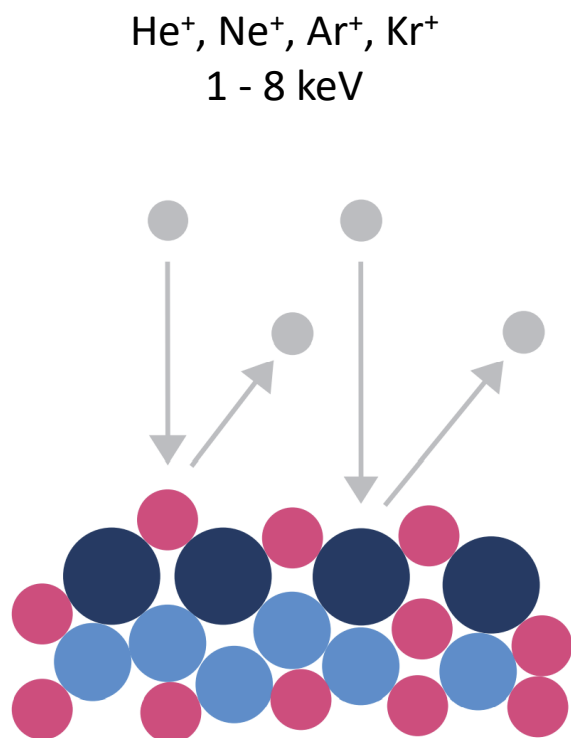
He⁺, Ne⁺, Ar⁺, Kr⁺
1 - 8 keV



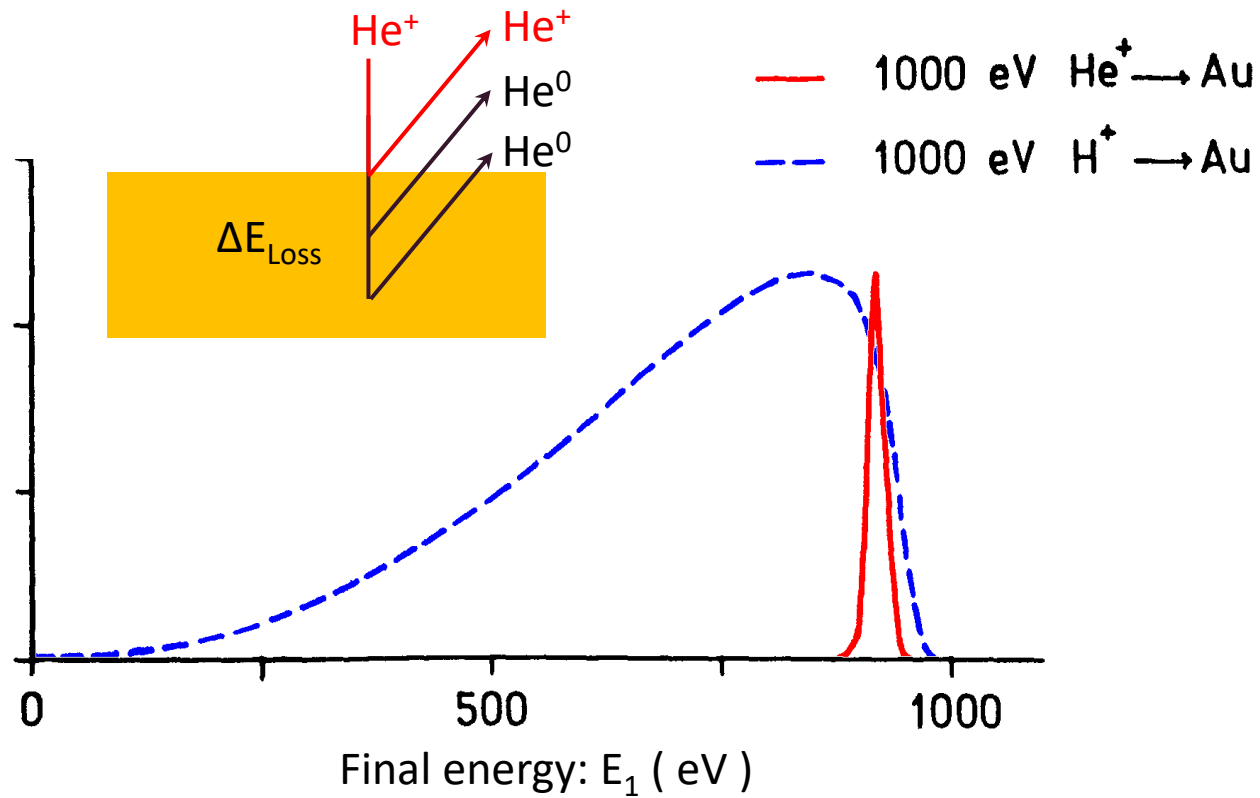
$$E_S = k \cdot E_P = \left(\frac{\cos\Theta + \sqrt{\left(\frac{M_S}{M_P}\right)^2 - \sin^2\Theta}}{1 + \frac{M_S}{M_P}} \right)^2 \cdot E_P$$

for $\frac{M_S}{M_P} \geq 1$

Energy of scattered ions (E_S) is following the laws of the **conservation of energy and momentum**

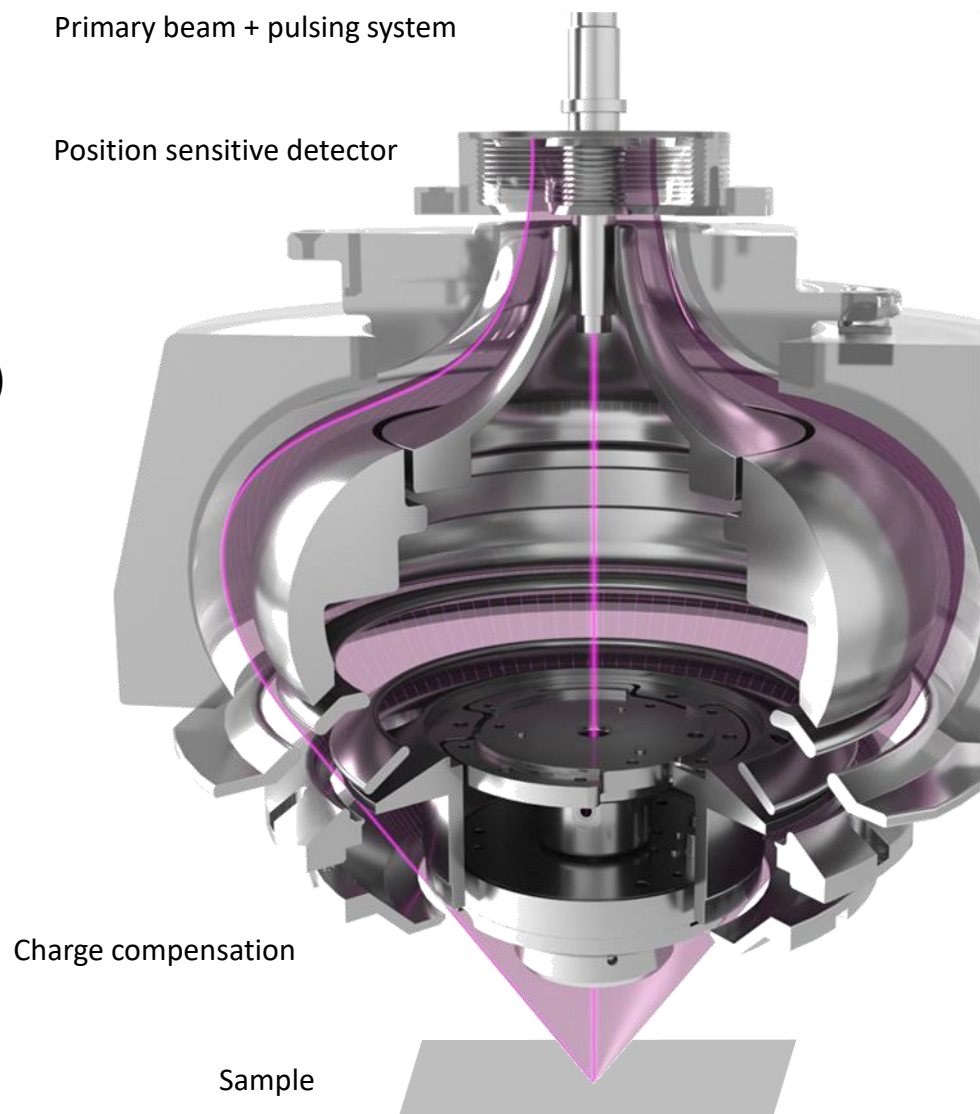


- > Ultra-high surface sensitivity, top atomic layer analysis
- > Static depth profiling information (up to 10 nm)
- > Reliable and straight-forward quantification
- > Simple analysis of insulators and rough samples
- > Detection of all elements > He
- > Detection limits (of 1 ML):
 - Li - O $\geq 1\%$
 - F - Cl 1% - 0.05%
 - K - U 500 ppm - 10 ppm

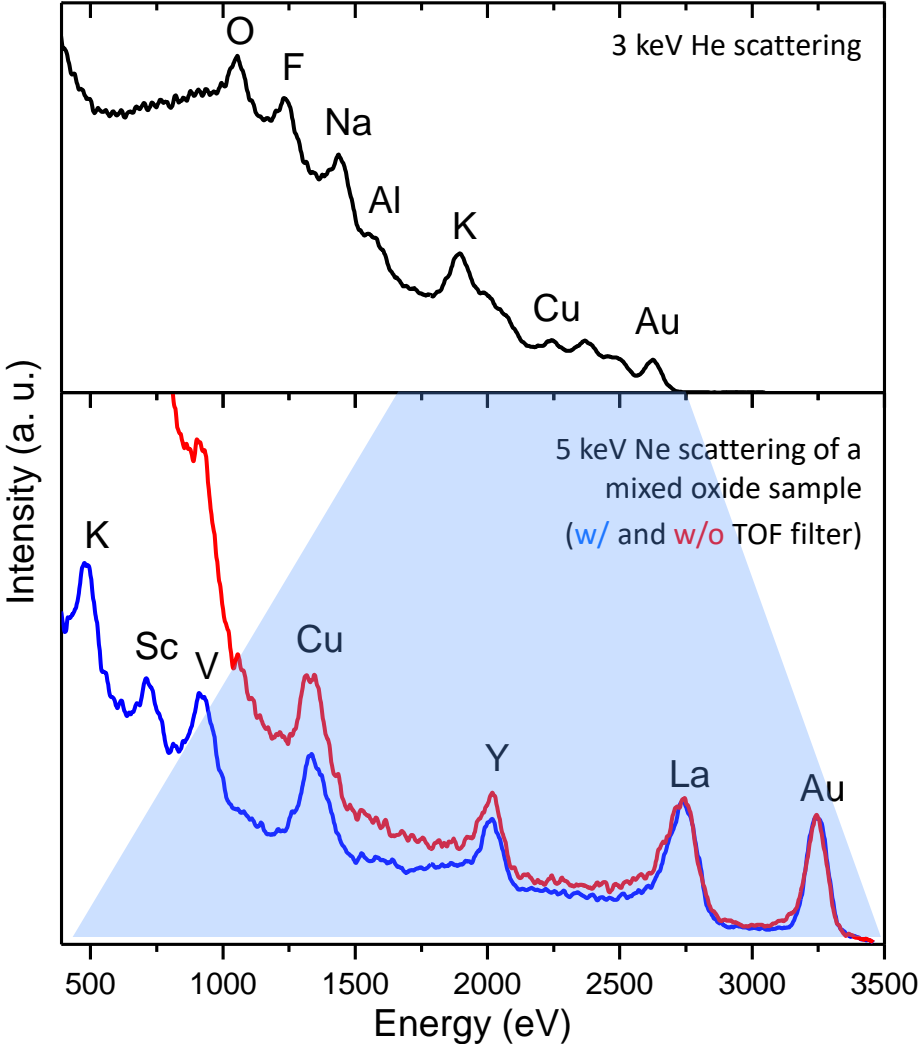
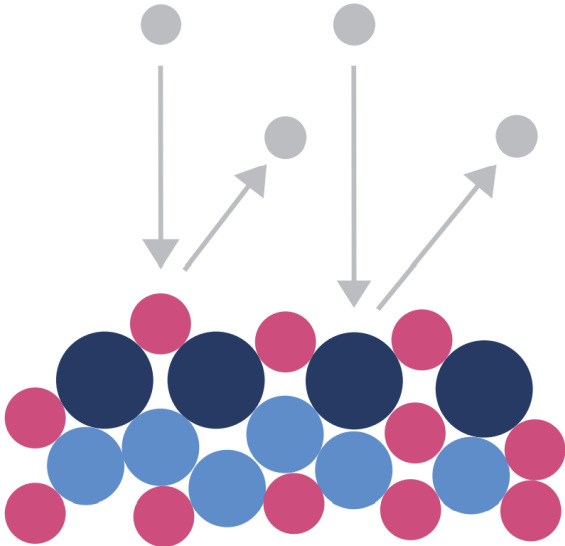


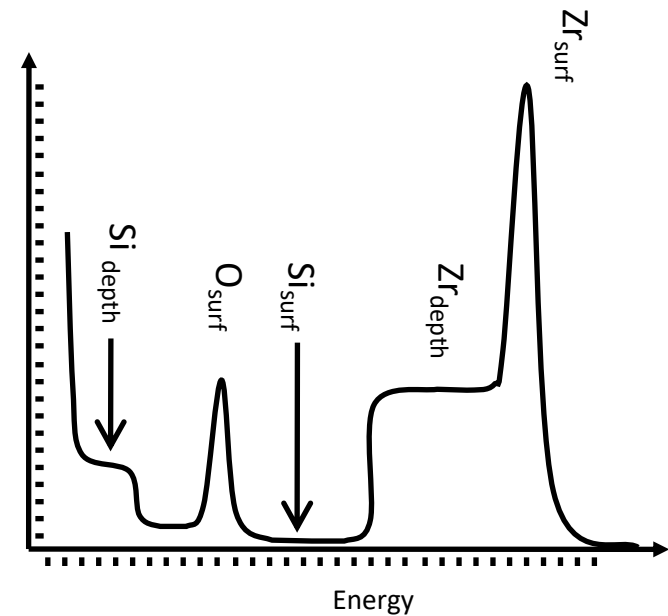
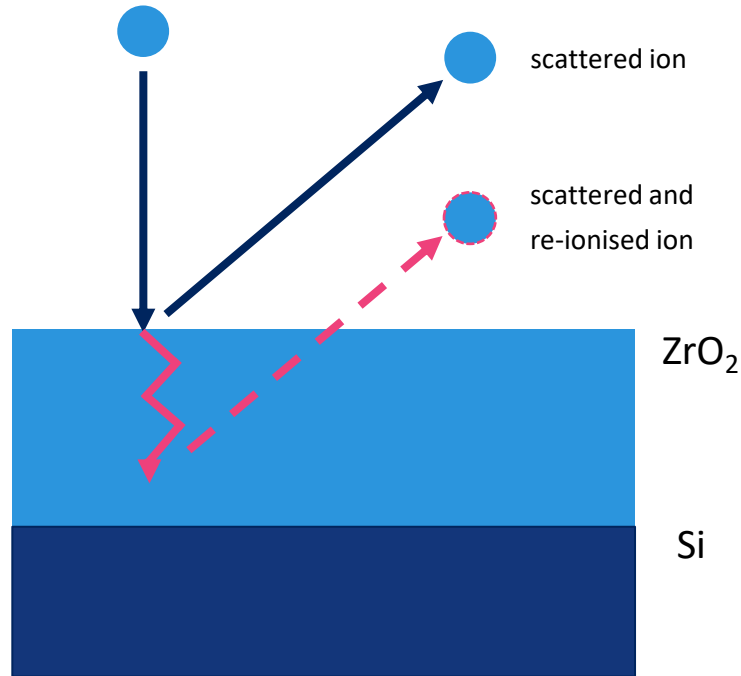
- > Neutralization effect of noble gas ions when penetrating the surface allows for extreme surface sensitivity

- > Crucial for practical work:
Analysis before destruction
- > High sensitivity and high-resolution analyzer:
 - ✓ Parallel detection of energy
 - ✓ Parallel acceptance of angles (azimuth)
 - ✓ Well defined scattering angle for high mass resolution

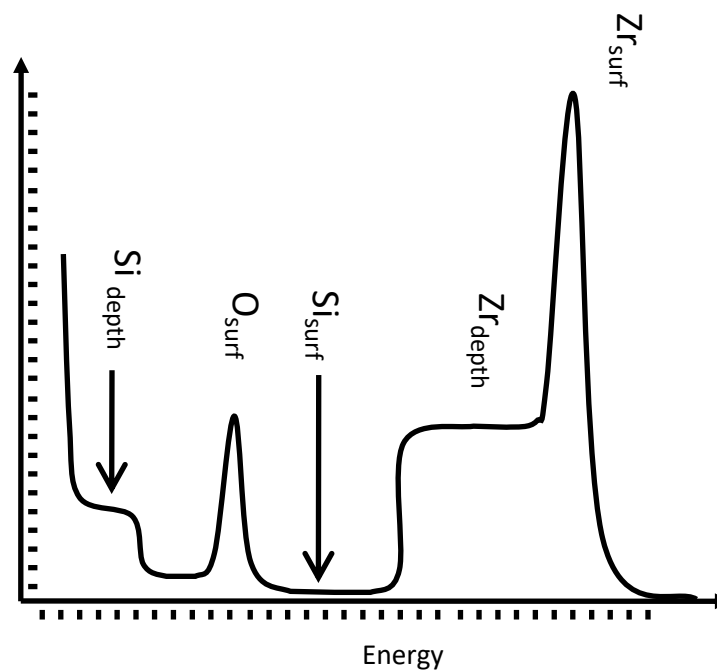
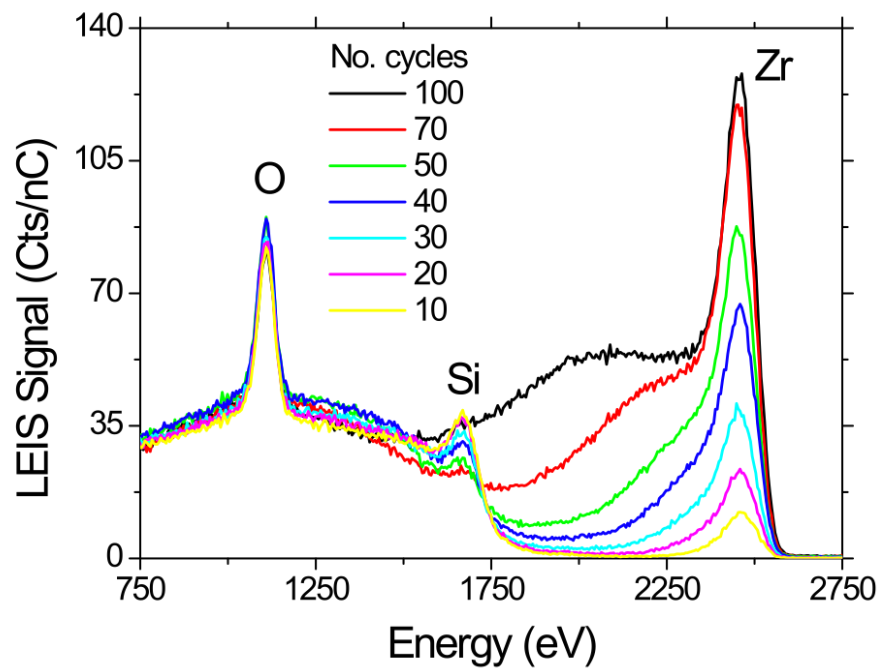


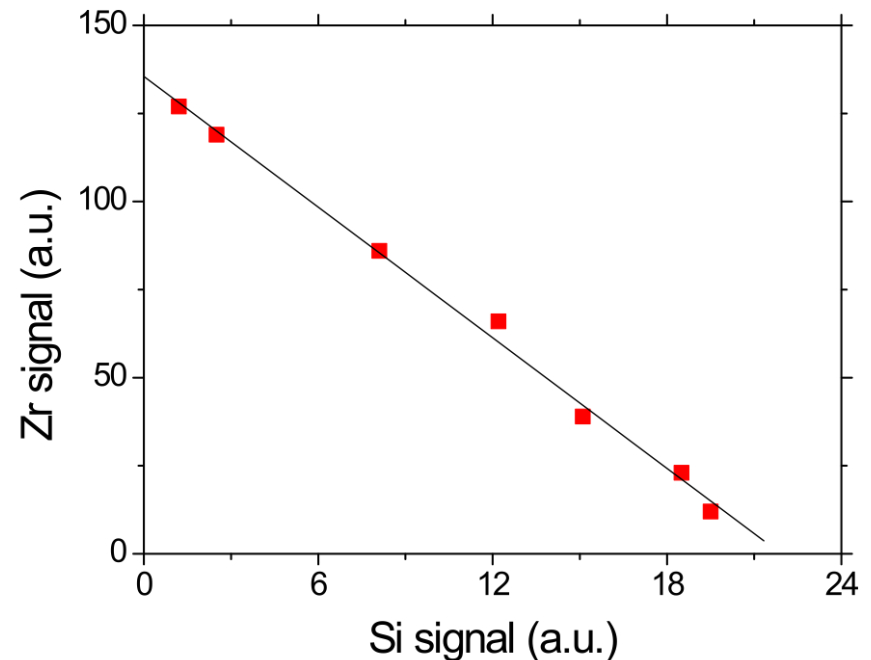
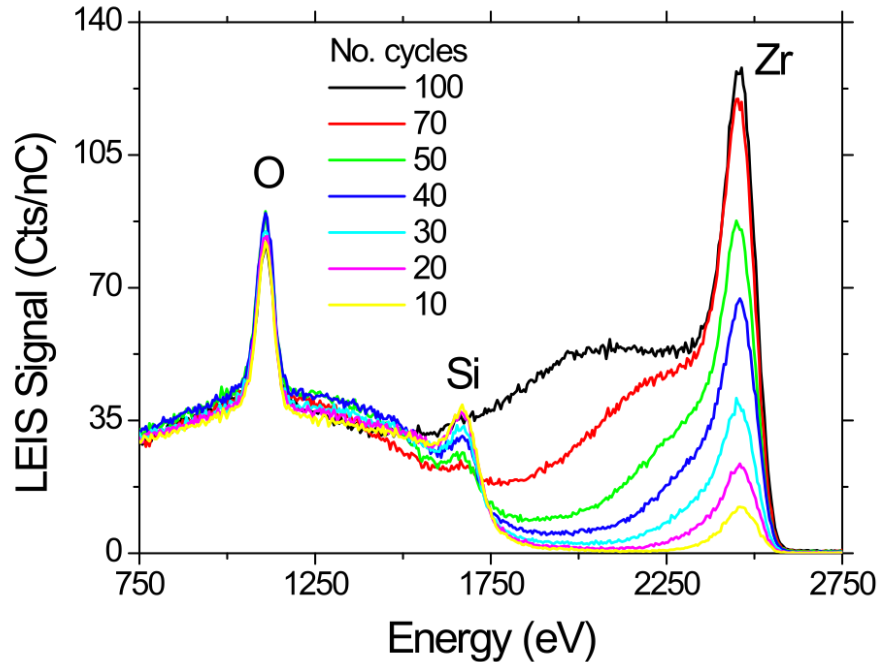
> Switching to Ne and Ar scattering expands mass scale for heavier elements



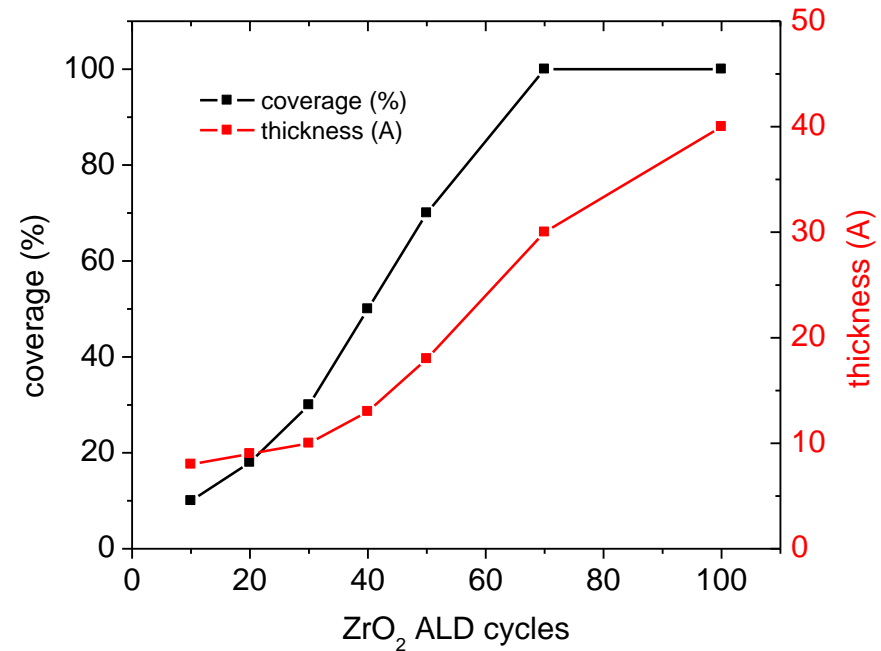
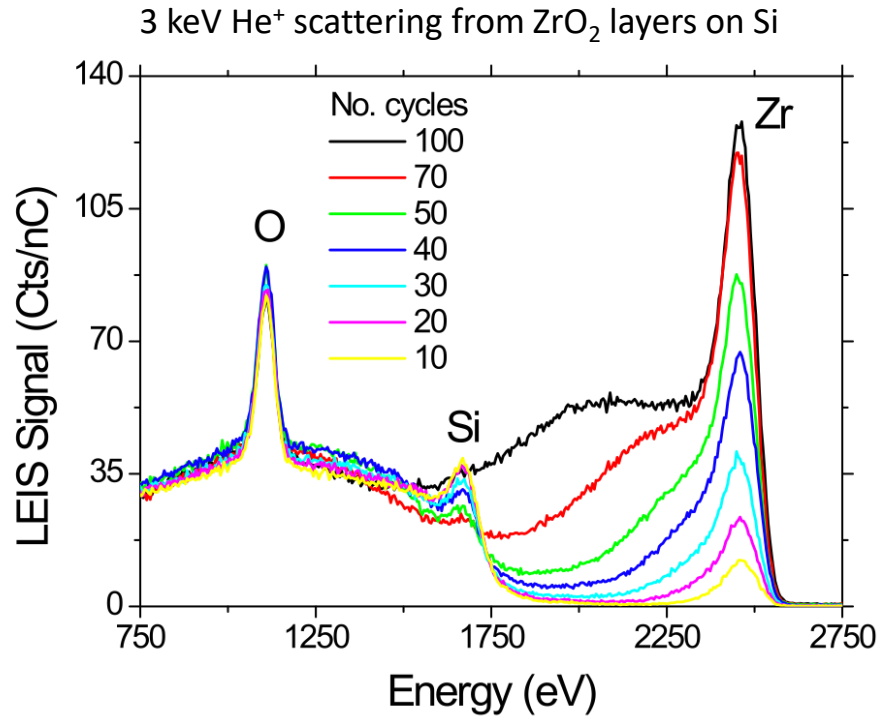


- > Ions can be scattered at the surface, giving element specific surface peak
- > Ions are also scattered in deeper layers, undergoing an additional energy loss proportional to the depth
- > When scattered in the volume, a re-ionization at the surface is required for detection. This is promoted by some elements (e. g. oxygen) and gives tails to the left of the peaks
- > Energy loss can be converted to depth





- Correlation plots: Extrapolation to both axes gives sensitivity factors for pure materials
- This allows reference free quantification in two component systems (and in many cases also with three components)



- Zr peaks develops a tail long before reaching maximum intensity (= layer closure)
- Coverage and thickness can be measured independently

Catalysis

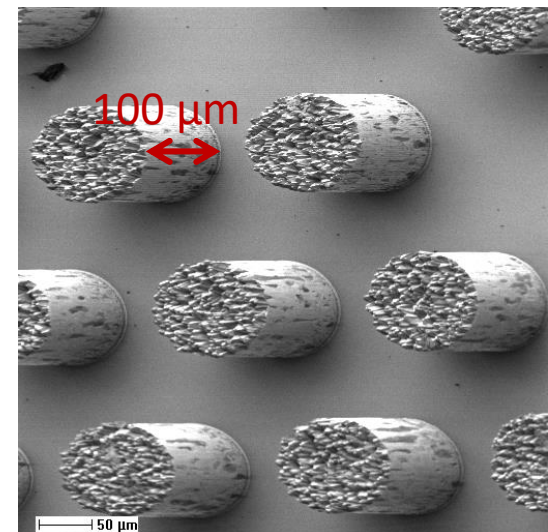
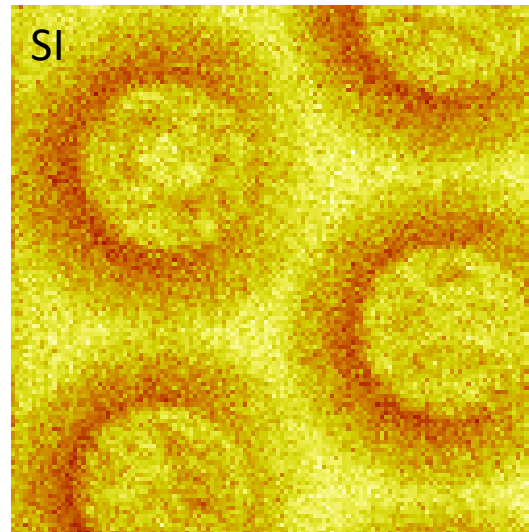
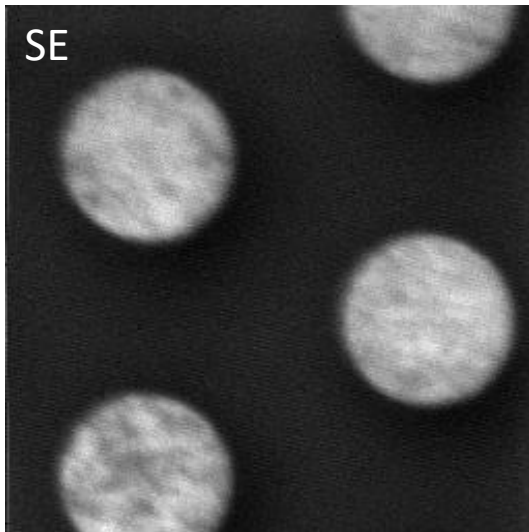
- > Analysis of catalysts with huge surface area routinely done
- > Depth of field of analyzer: approx. 500 μm

Table 2. Relative LEIS yields of a flat quartz sample and pressed silica powders (300 MPa) with specific surface areas of 50–380 $\text{m}^2 \text{g}^{-1}$

Sample	Specific surface area ($\text{m}^2 \text{g}^{-1}$)	Si	O	Si/O ratio
Quartz	$\sim 10^{-4}$	1.00	1.00	1.00
Aerosil Ox-50	50	0.80	0.84	0.95
Aerosil 130	130	0.77	0.81	0.95
Akzo Düren	185	0.83	0.82	1.01
Aerosil 380	380	0.83	0.84	0.99

Jansen et al., Surf. Interface Anal. **36** (2004) 1469

FoV: 300 x 300 μm^2



> Pressure gap

- Pressure inside the reactor: ≈ 10 bar
- Pressure inside the analysis chamber: $\approx 10^{-6} - 10^{-10}$ mbar

> Structure gap

- Low loading, rough surfaces ($1000 \text{ m}^2/\text{g}$) inside the reactor
- High loading and flat surfaces inside the analysis chamber

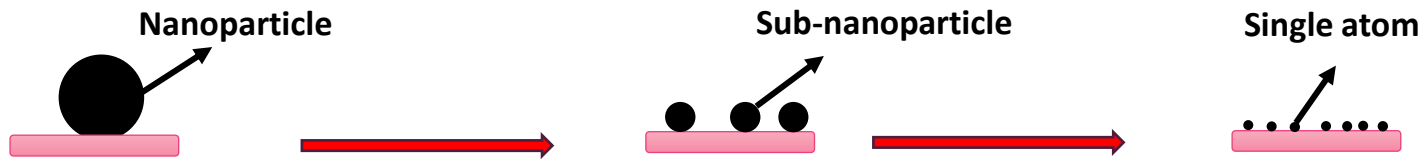
> The Qtac bridges the structure gap but the pressure gap exists partially.

> Workaround

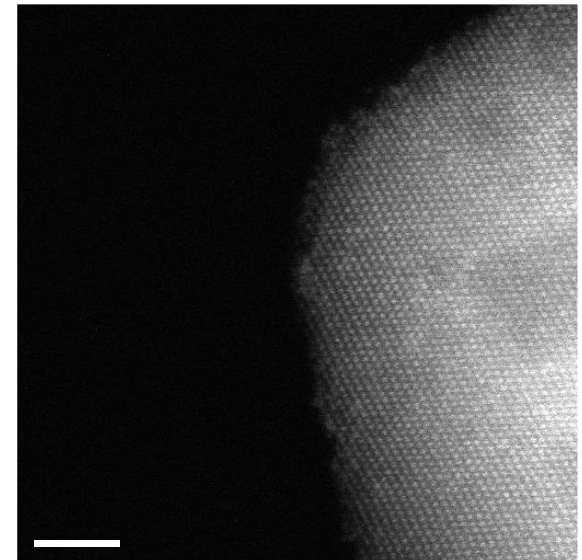
- In-situ preparation of the catalyst with subsequent quenching

Single atom heterogeneous catalysis

- > improved efficiency, higher reactivity, and better selectivity
- > lower loading of precious metals

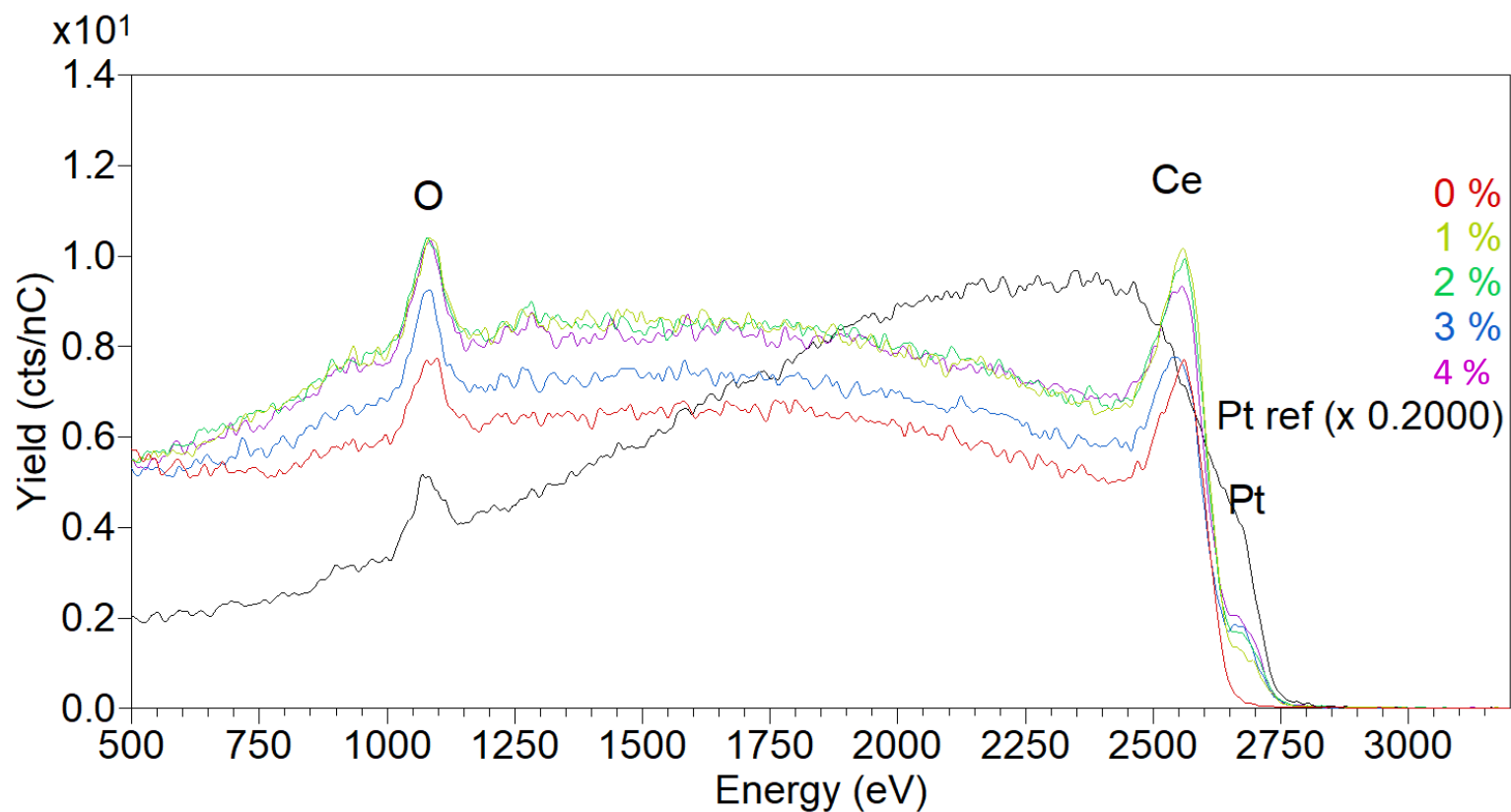


- > Pt/CeO₂, prepared using atom trapping
- > 1, 2, 3, 4 wt. % Pt, 10 h @ 800° C in air
- > At higher loading, large Pt particles expected
- > Small Pt particles are not stable at elevated T, evaporation of PtO₂
→ either single atoms or large particles

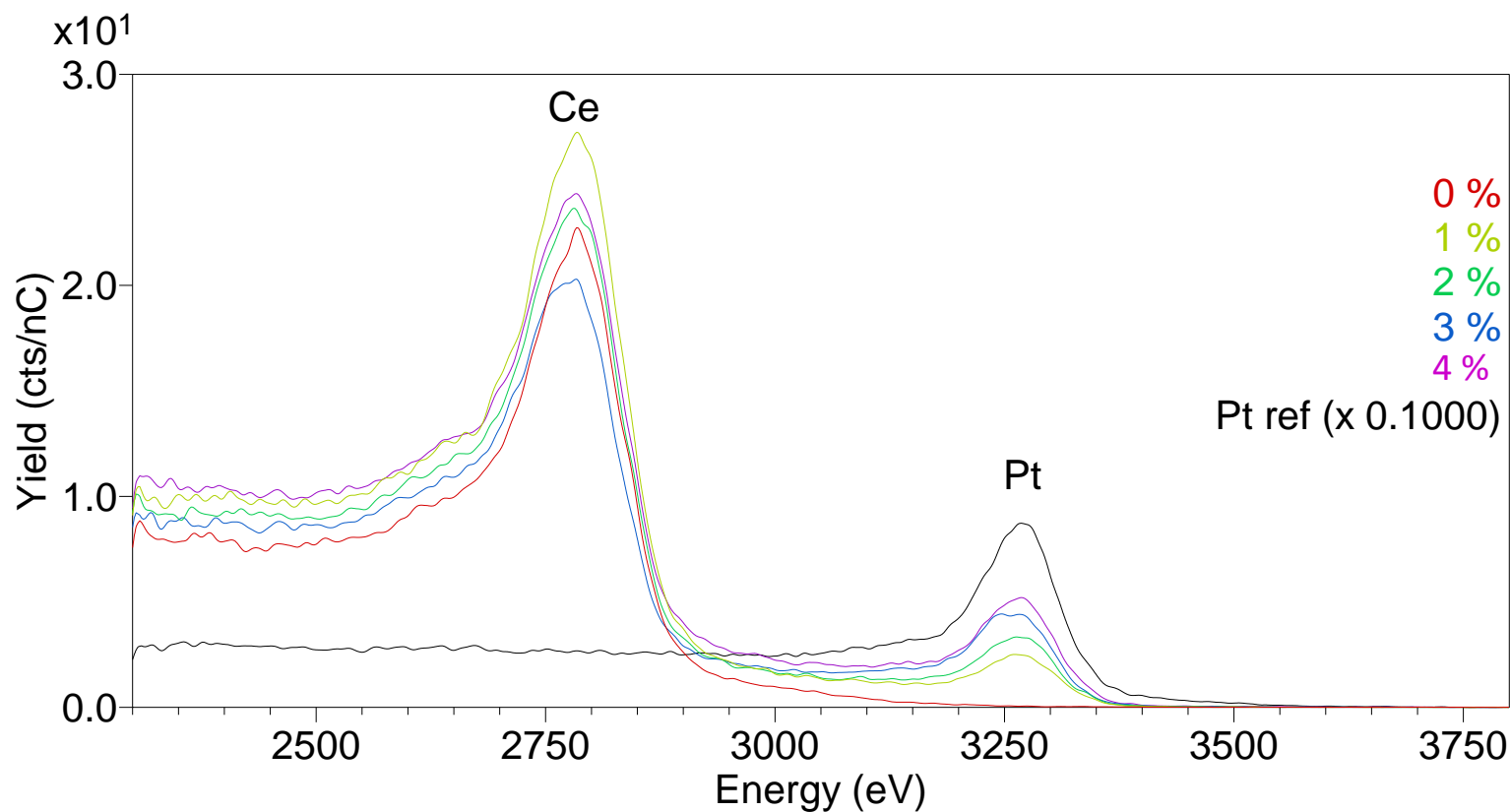


Jones et al., Science (2016)

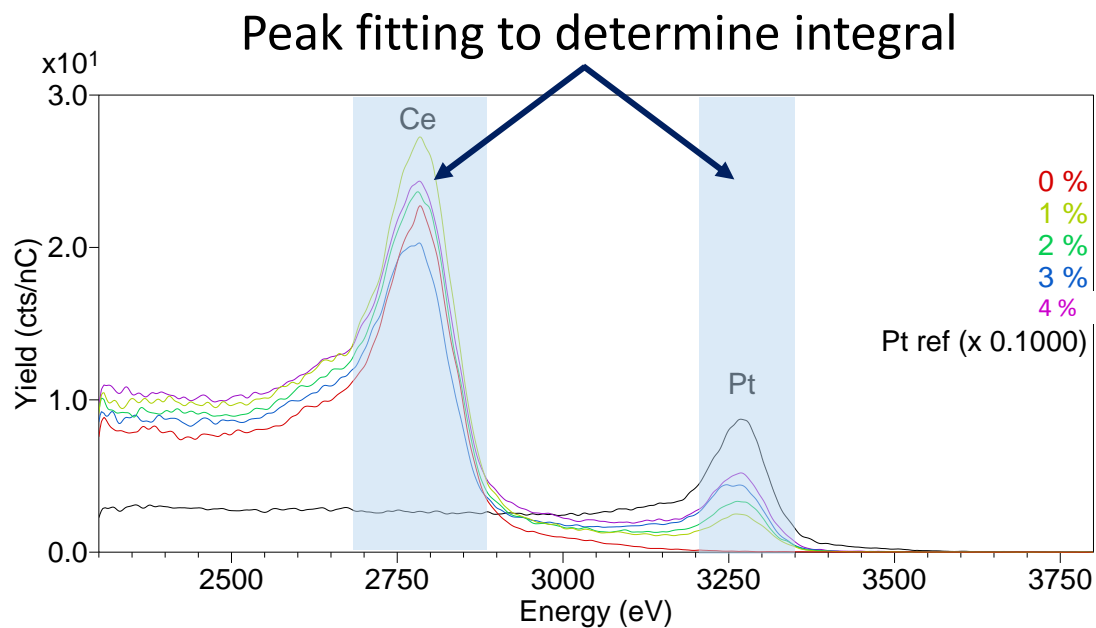
- > Samples and Pt metal cleaned using atomic oxygen
→ no C, organics
- > No unexpected elements at the surface
- > Mass resolution not sufficient for Ce/Pt → Ne scattering



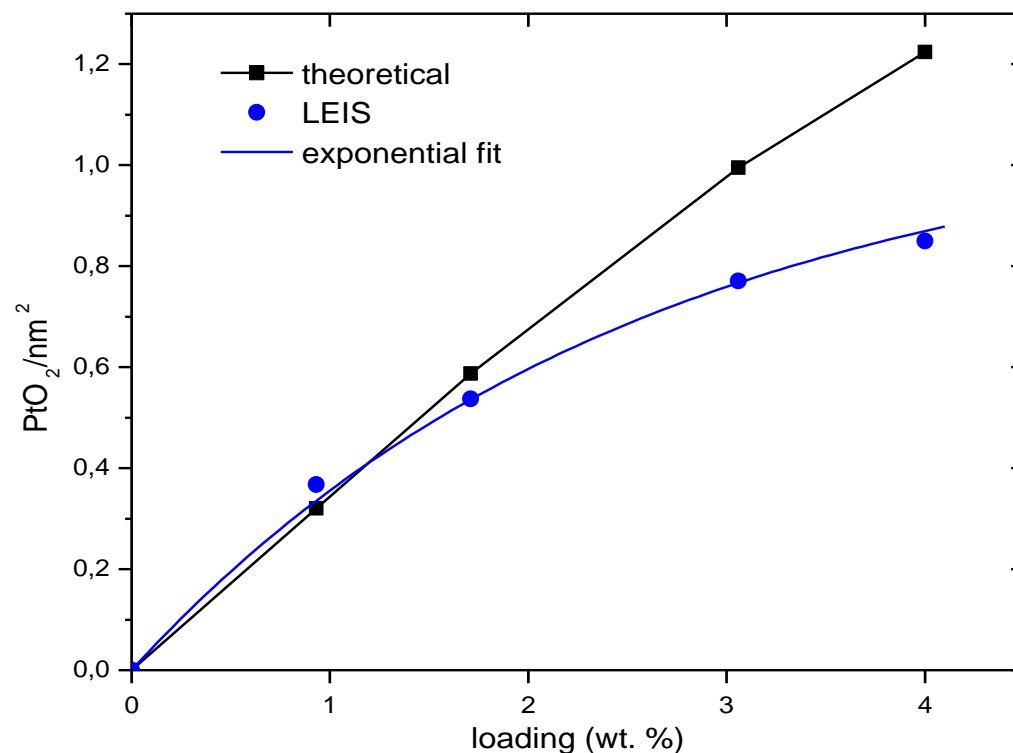
- > Excellent mass resolution
- > Quantification by comparison to Pt reference



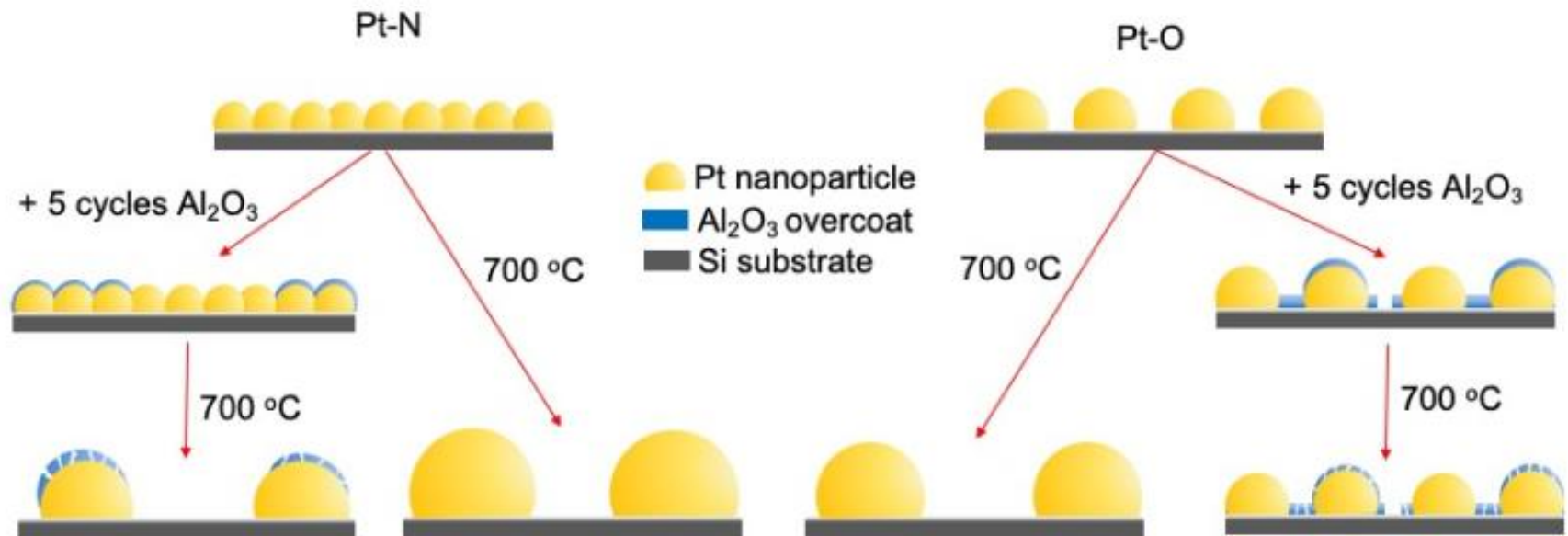
- > Determine LEIS signal for reference PtO₂ – 6820 cts/nC
- > Calculate PtO₂ density of reference: 9.01 PtO₂/nm²
- > Calculate PtO₂ density of samples: $Y_{\text{LEIS}}/6820 * 9.01 \text{ PtO}_2/\text{nm}^2$
- > Apply small roughness correction – catalyst and reference are very different



- > At 1 and 2 wt. %, Pt is present as single atom catalyst – quantitative agreement, no normalization
- > At 3 % loading, 77 % of the atoms are detected in the outer layer
- > At 4 % loading, 69 % of the atoms are in the outer layer
- > As large particles only minimally contribute to the surface, their signal is weak in LEIS

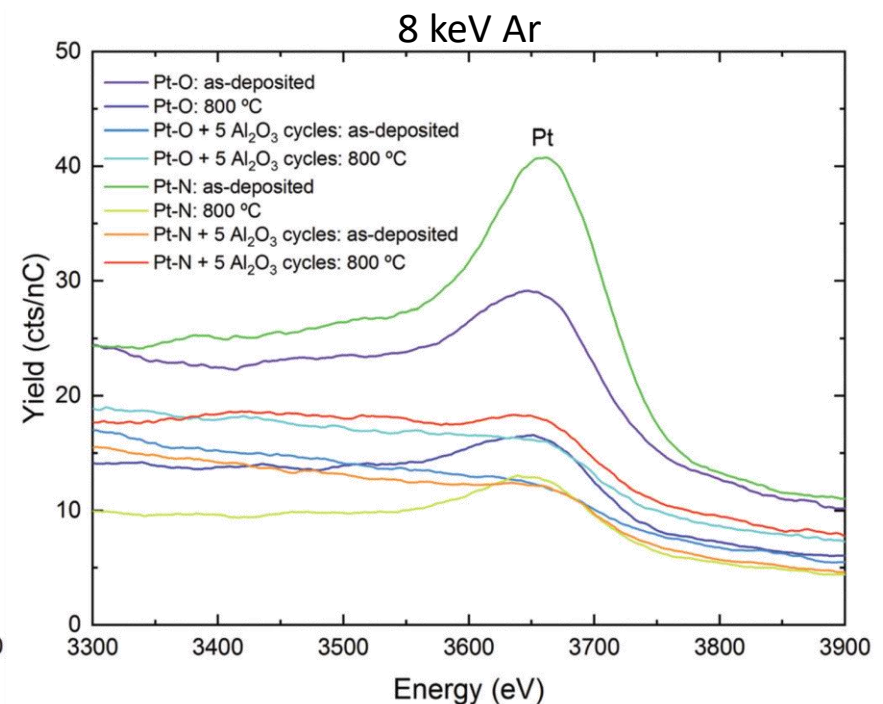
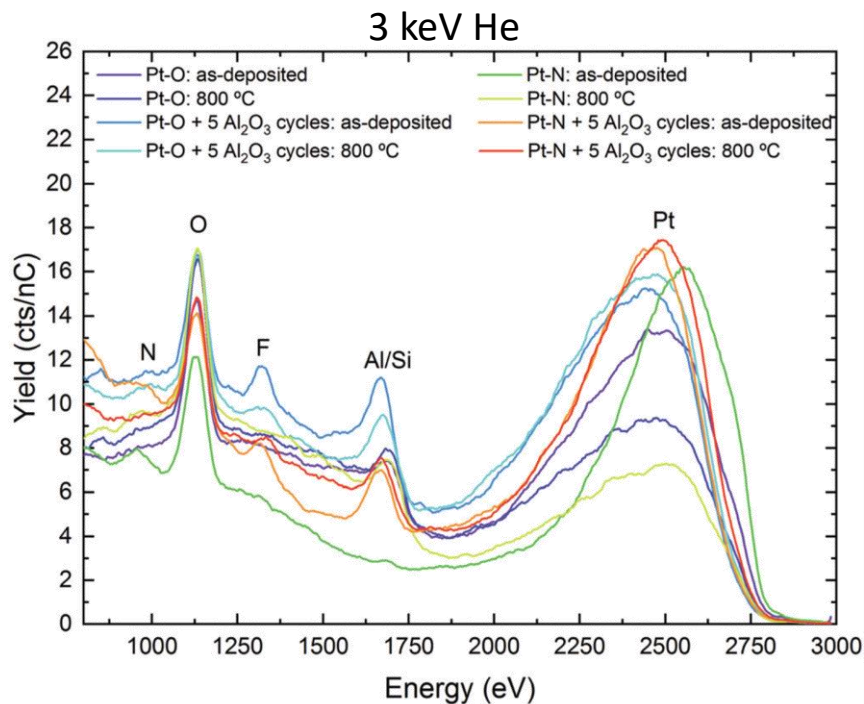


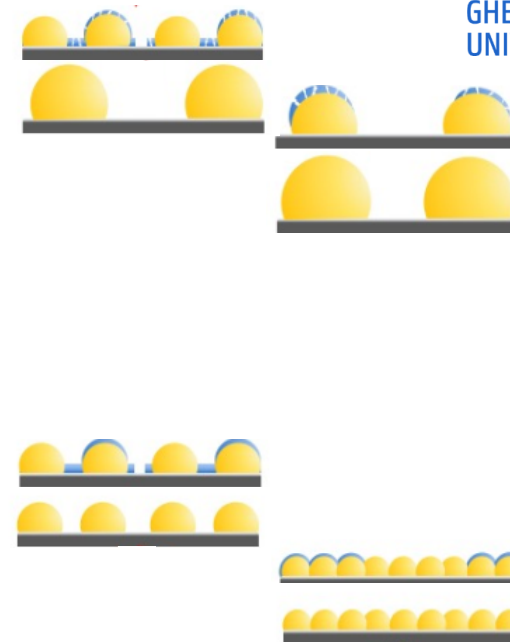
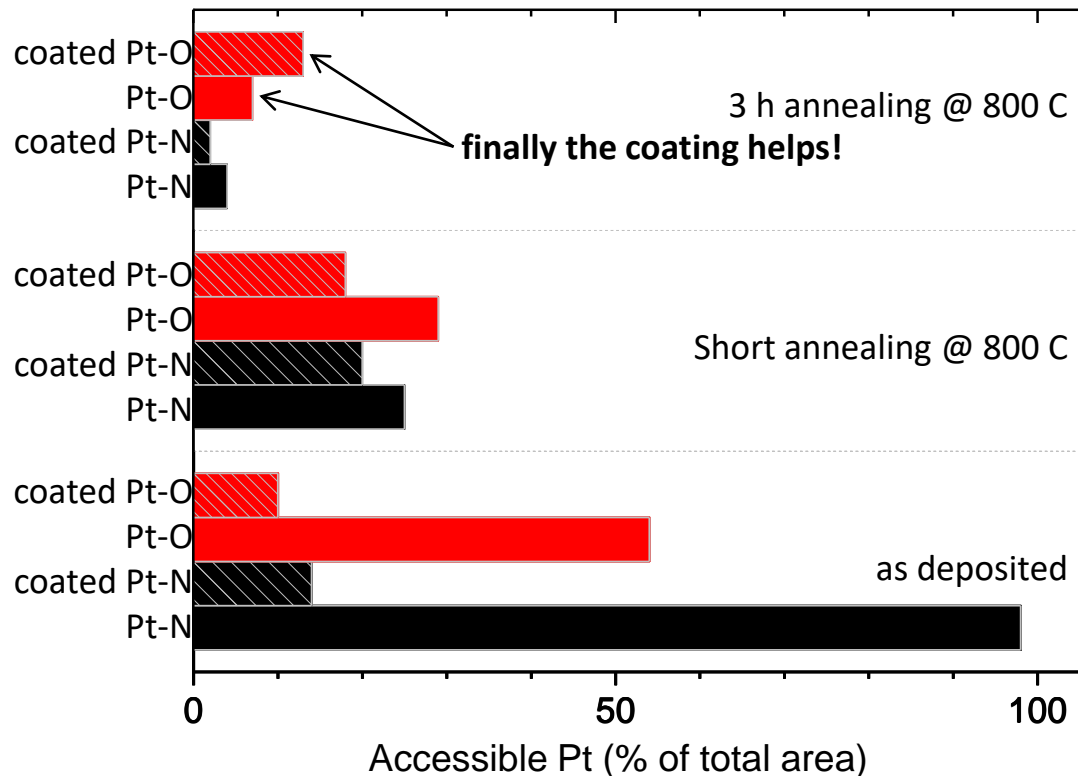
- > Full picture from multiple techniques





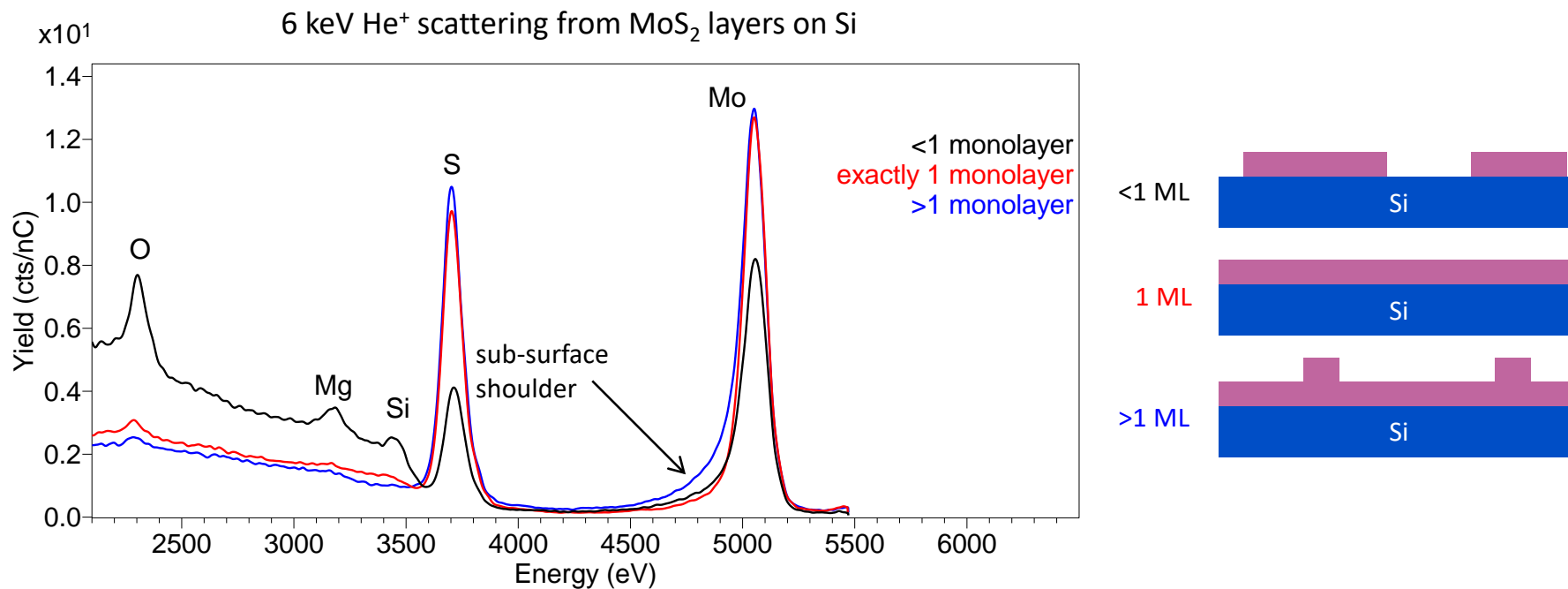
- > Nanoscale Pt particles are desirable for catalytic activity **and** efficiency
- > Problem: Particle coarsening due to harsh thermal and chemical conditions during catalysis
- > Idea: ALD Al₂O₃ overcoating to prevent particle coarsening





- > XRF (RBS calibrated) quantifies total Pt amount
- > in-situ GISAXS measures particles coarsening
- > LEIS quantifies availability of Pt even after Al₂O₃ overcoat
- > Key result: Isolated particles are required to prevent coarsening by ALD

Ultra-thin films



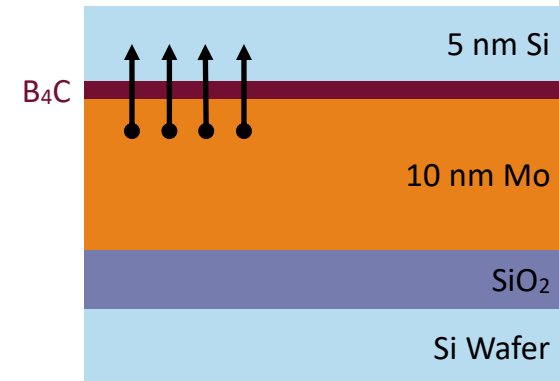
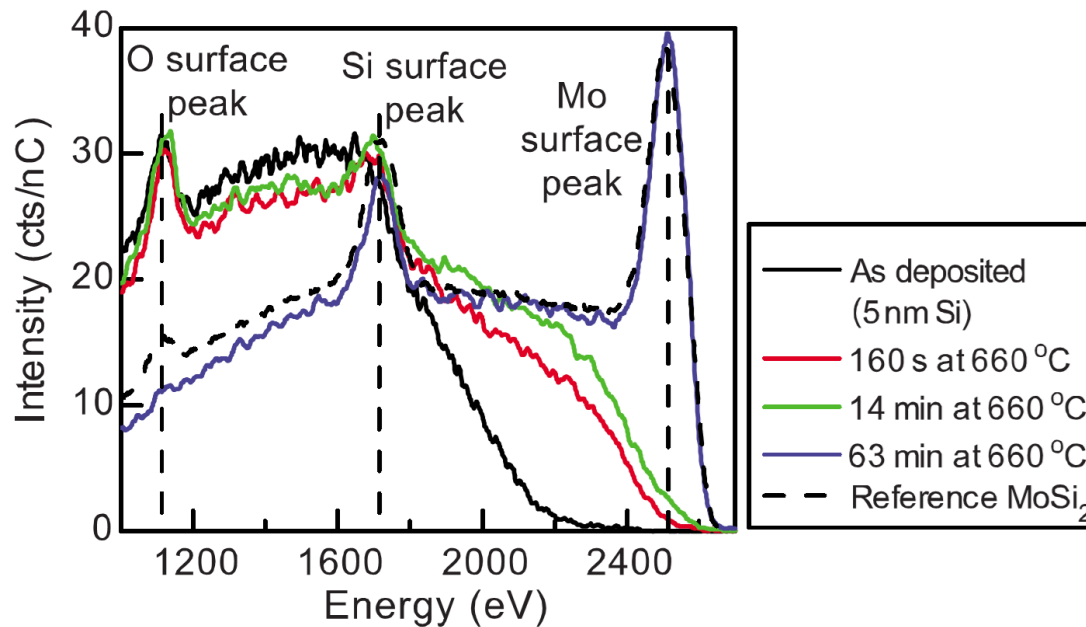
> Mo surface peak:

- Symmetric Gaussian: (sub-)monolayer coverage
- Peak integral proportional to fraction of surface covered
- Tail for >1 monolayer developing

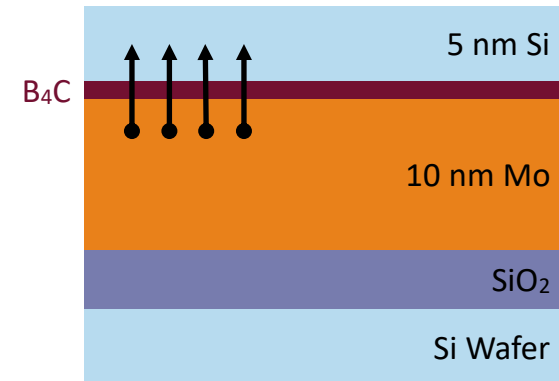
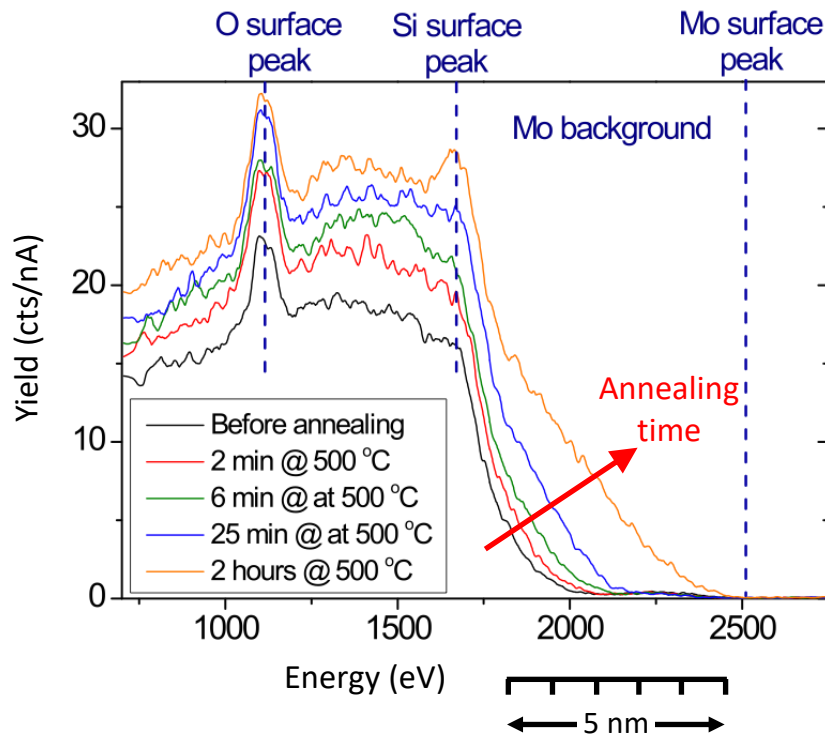
Samples courtesy of Jeong-Gyu Song



- > 5 nm Si / 1.6 nm B₄C / 10 nm Mo, annealing @ 660 deg. C



- > 5 nm Si / 1.6 nm B₄C / 10 nm Mo, annealing @ 500 deg. C
- > Diffusion coefficient without B₄C : $(8 \pm 2) \cdot 10^{-20} \text{ m}^2/\text{s}$
- > Diffusion coefficient with 1.6 nm B₄C : $(4 \pm 1) \cdot 10^{-21} \text{ m}^2/\text{s}$



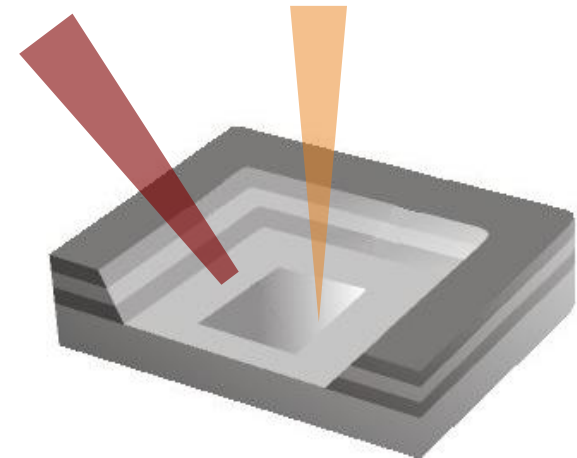
V. de Rooij-Lohmann, *Appl. Phys. Lett.* **94** 063107 (2009)

V. de Rooij-Lohmann, *J. Appl. Phys.* **108** 014314 (2010)

- > Sputter and analysis beam conditions are optimised independently
- > Sputtering using inert species (usually Ar) at low energy to assure high depth resolution
- > Scattering using a noble gas ion beam selected for optimum sensitivity and mass resolution for the elements of interest

Ar, Kr, Xe, ...

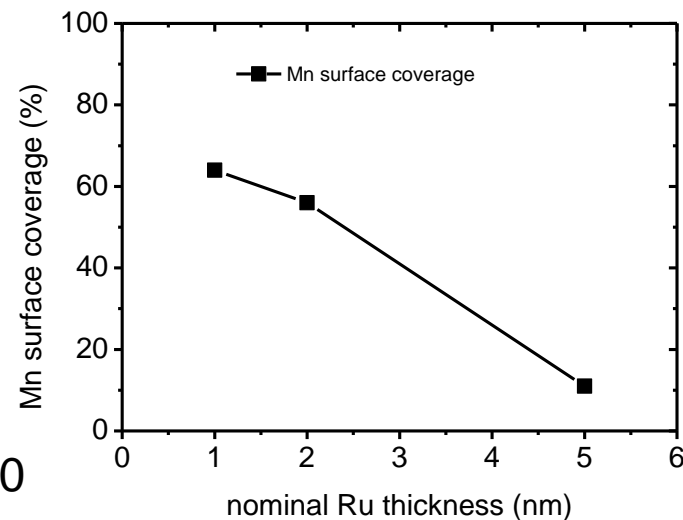
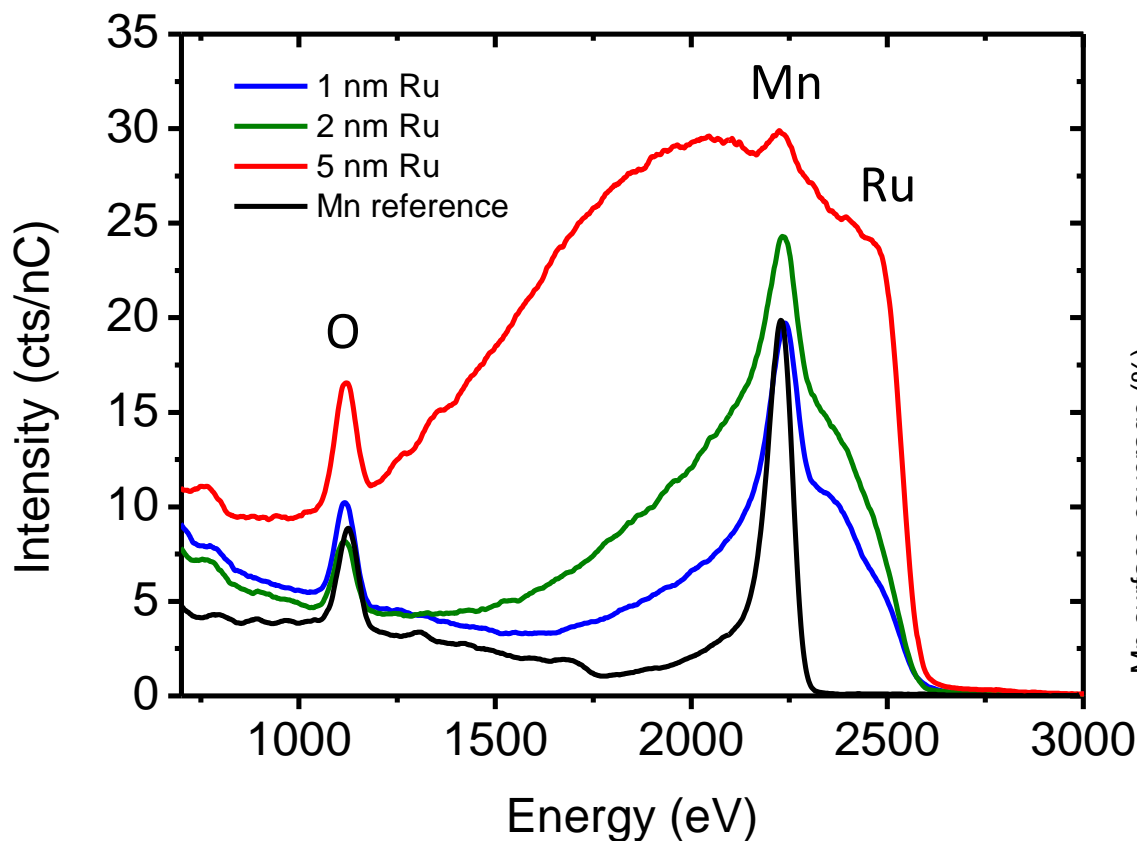
^3He , ^4He , Ar, ...



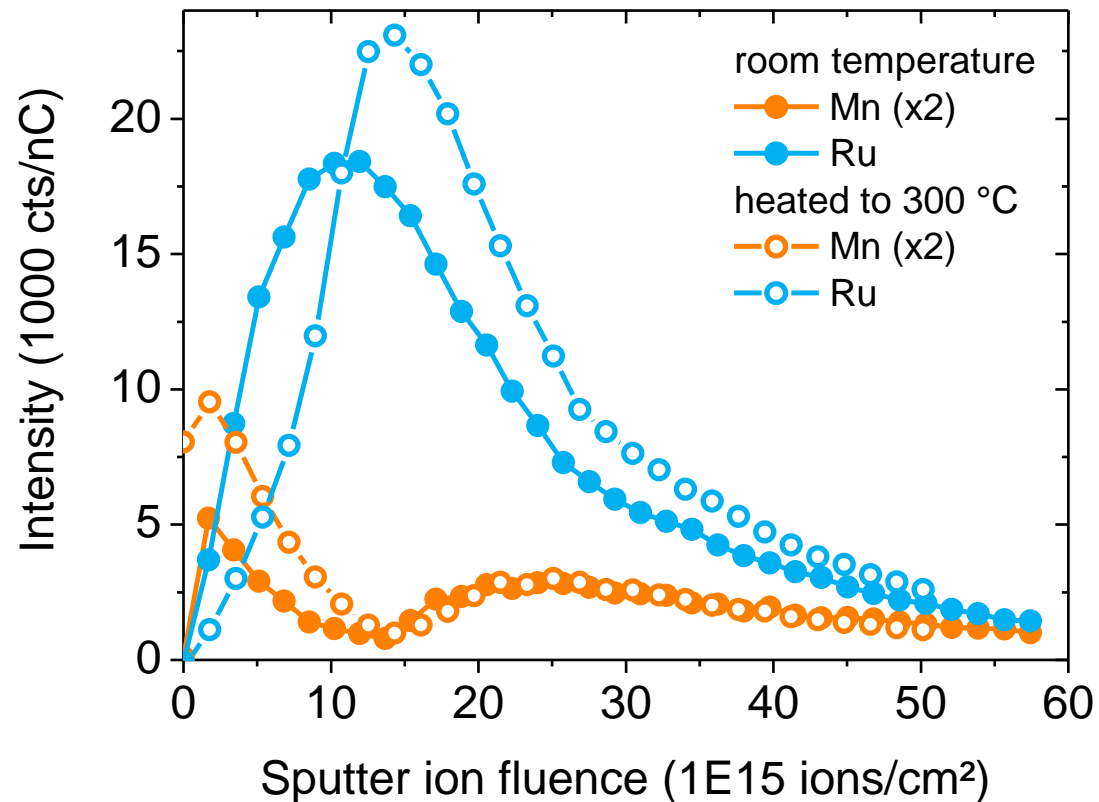
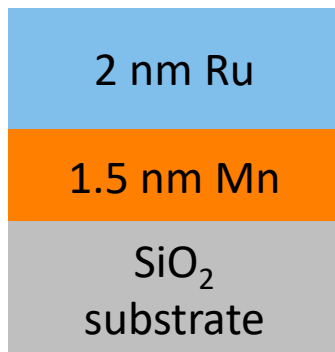
Sputter area: $2 \times 2 \text{ mm}^2$

Analysis area: $1.5 \times 1.5 \text{ mm}^2$

- > 1/2/5 nm Ru on 1.5 nm Mn on Si
- > Ru surface peak shielded after atomic H treatment
- > Mn visible despite a layer of Ru covering it, even at 5 nm Ru – why?

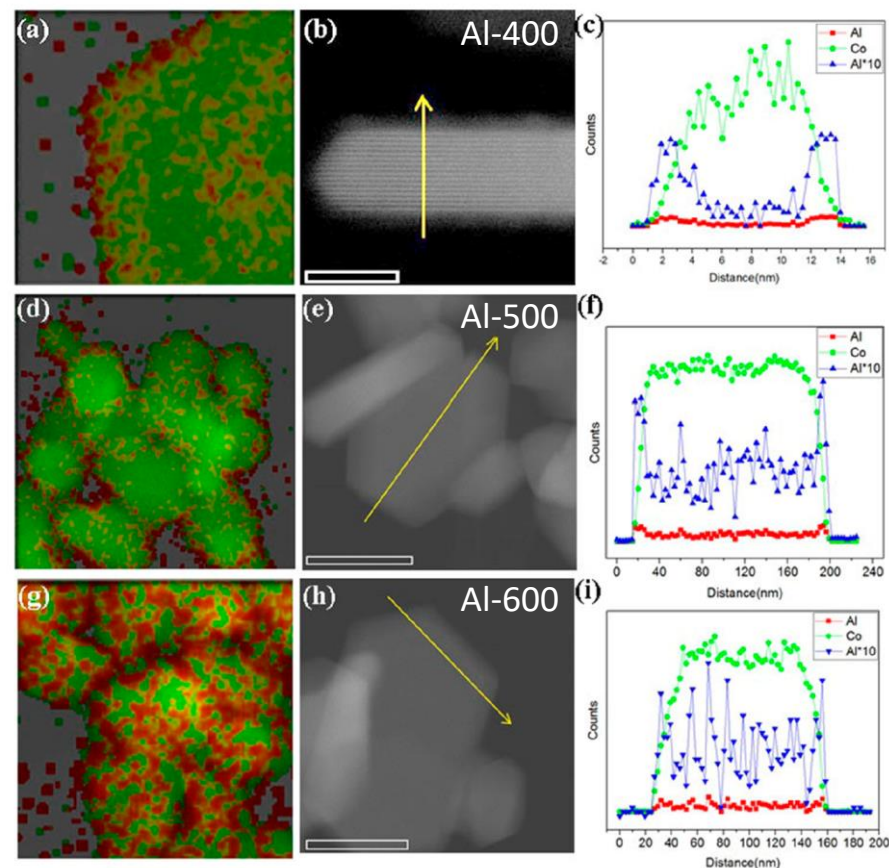


- > 500 eV Ar sputter profile before and after heating to 300 °C
- > Mn is enriched at the surface and below the Ru
- > Enrichment increased by heating → diffusion through Ru film



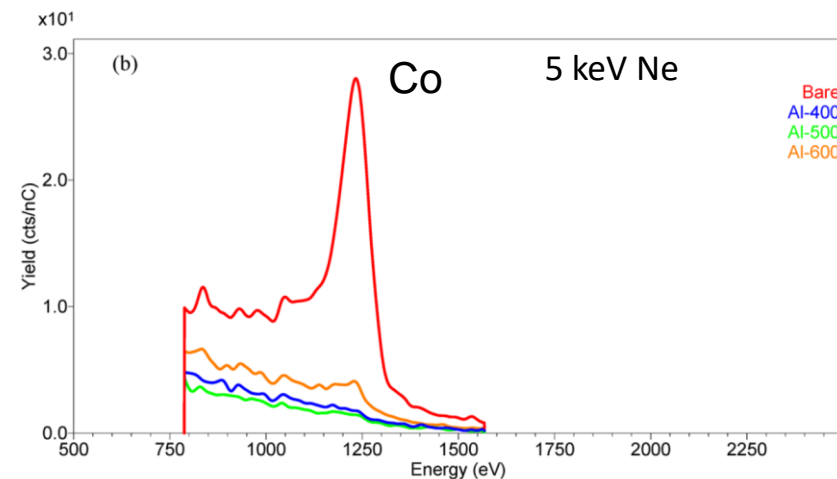
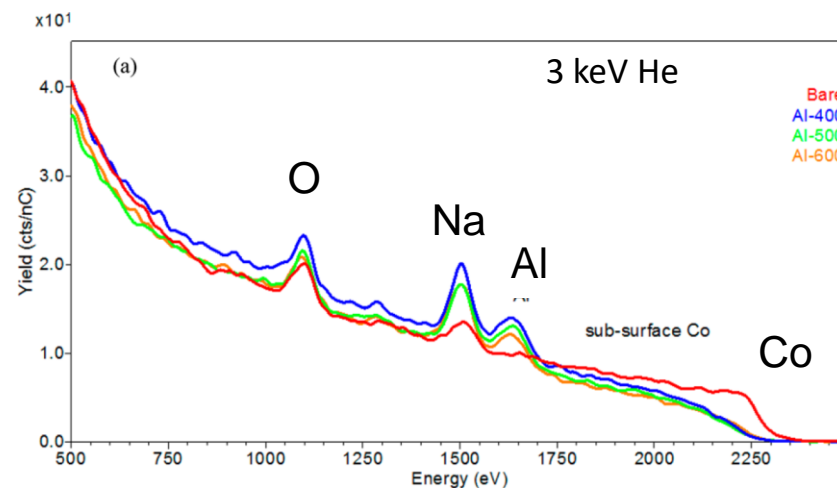
Energy materials

- > Li transparent passivation coating on electrode surfaces
- > Wet chemical Al_2O_3 on LiCoO_2 nano-platelets
- > Calcination at 400°C , 500°C , 600°C for 3 h (Al-400, Al-500, Al-600)
- > EDX mapping shows coating, diffusion of Al present into core
- > Quantification of surface coverage impossible
- > XRD also sees diffusion with T



EDX mapping (10 nm scalebar)

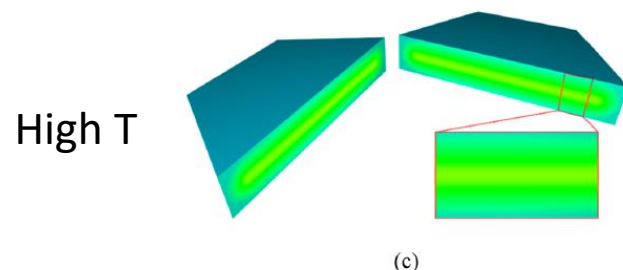
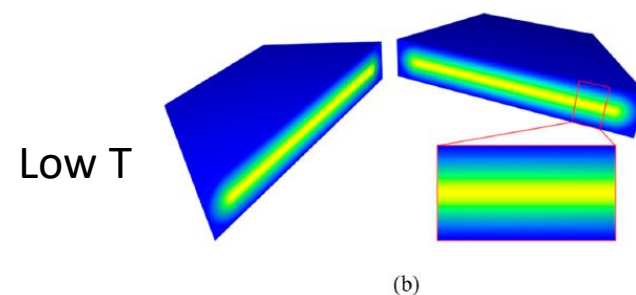
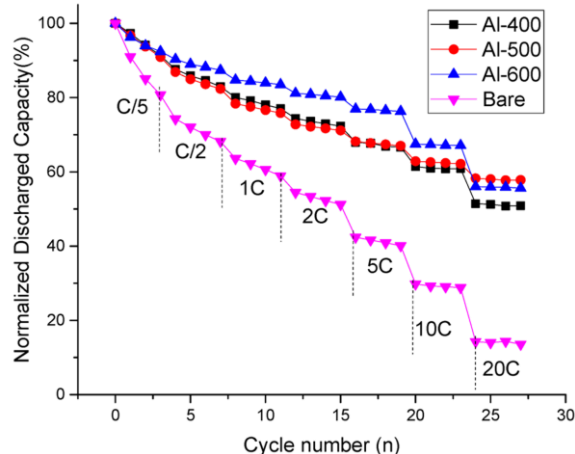
- > LEIS detects Al and residual Co at the surface (incomplete coating/diffusion)
- > Sub-surface Co hardly changing
- > Several 10 % Na detected (wet chemistry)



sample	Co signal (cts/nC)	Co surface coverage (%)
Bare	2208	100
Al-400	41	2
Al-500	41	2
Al-600	199	9

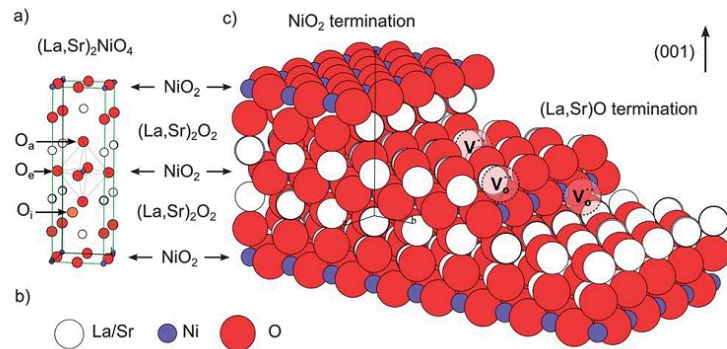
- > Calcination stimulates diffusion processes
- > No complete intermixing (otherwise Co surface coverage would be much higher in LEIS)
 - Al ox preferred at surface, also seen in LEIS on CoAl_2O_4

- > At similar coulombic efficiency, normalised discharge capacity best for Al-600

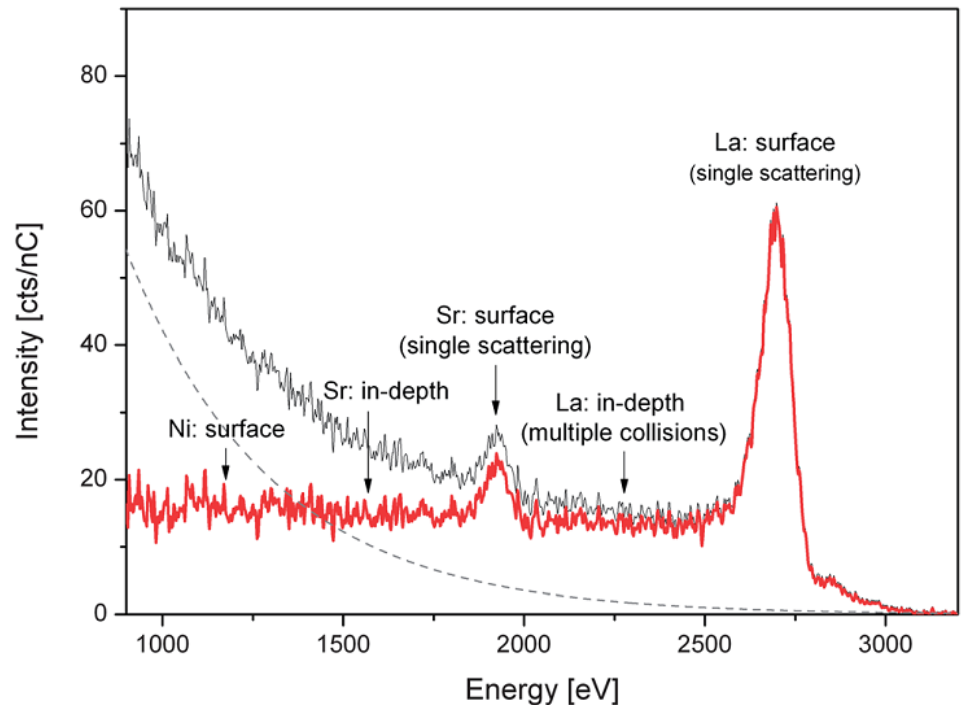


Absence of Ni on outer layer of Sr doped La_2NiO_4 single crystals

- > $(\text{La, Sr})_2\text{NiO}_4$ is a candidate for SOFC cathodes – ionic O conductor
- > Authors use LEIS, CTR and angle resolved XPS to analyze low index faces of as-is and heat treated crystals (450°C , 72 h in air)
- > LEIS data shows surface termination



Possible terminations of $(\text{La, Sr})_2\text{NiO}_4$

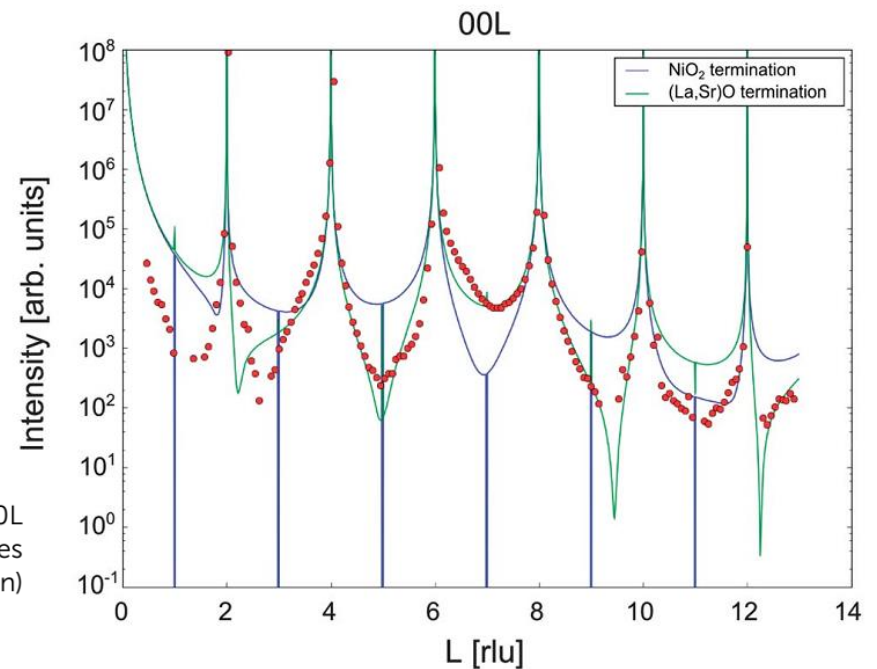


- > Angle resolved XPS and CTR less surface sensitive
- > Agreement with LEIS findings: no Ni in (110) and (001) w/ and w/o annealing

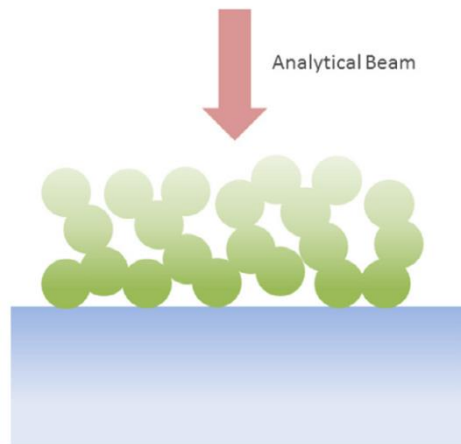
AR-XPS: (La+Sr)/Ni ratio vs. depth

Depth (nm)	As cleaved single crystal	Heat treated single crystal
0.6	5.3 ± 1.2	6.7 ± 1.9
1.8	8.5 ± 1.6	5.5 ± 1.1
3.5	6.5 ± 1.0	5.2 ± 0.9
4.8	3.9 ± 0.5	4.3 ± 0.7
5.9	3.4 ± 0.4	4.6 ± 0.6
6.6	3.1 ± 0.4	4.7 ± 0.5
7.0	2.3 ± 0.2	4.5 ± 0.4
Bulk (theoretical)	2.0	2.0

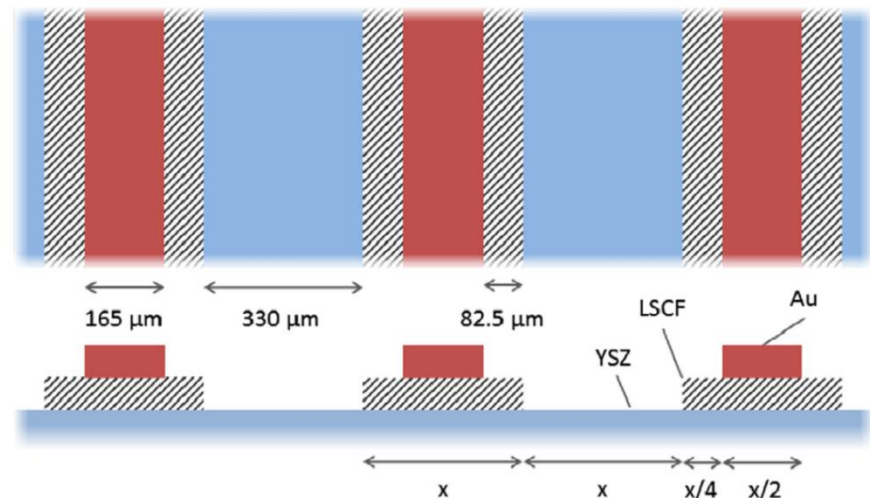
Fig. 2 Crystal truncation rod scattering as measured along the 00L direction in air at 450 °C. Red dots indicate the raw data and solid lines the model CTR patterns for NiO_2 (blue) or (La,Sr)O (green) terminations.



- > Fuel cell performance limited by oxygen exchange between solid electrolyte and gas phase
- > Kinetics determined by transport properties and surface chemistry
- > Interface usually not accessible to surface analysis → model structure

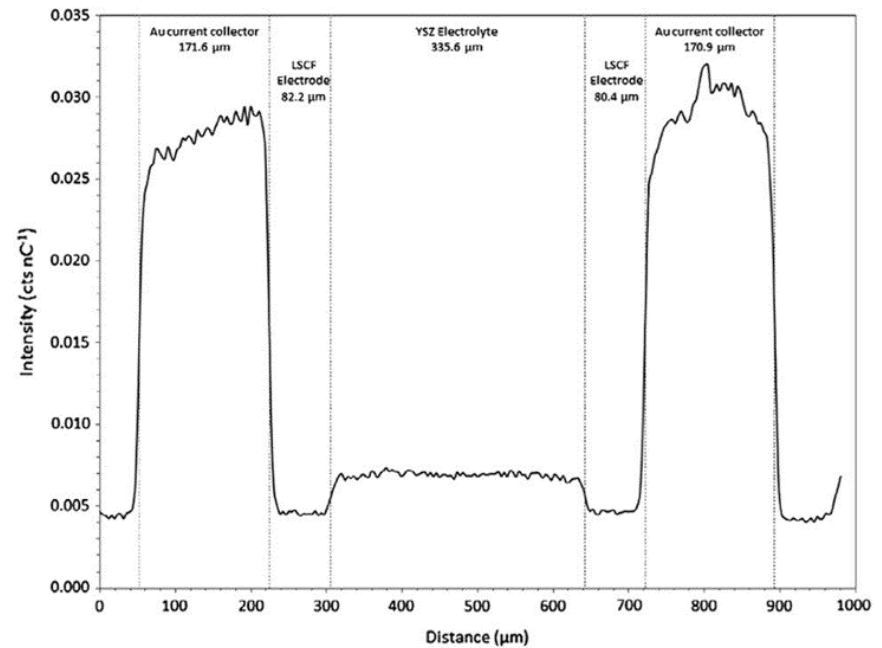
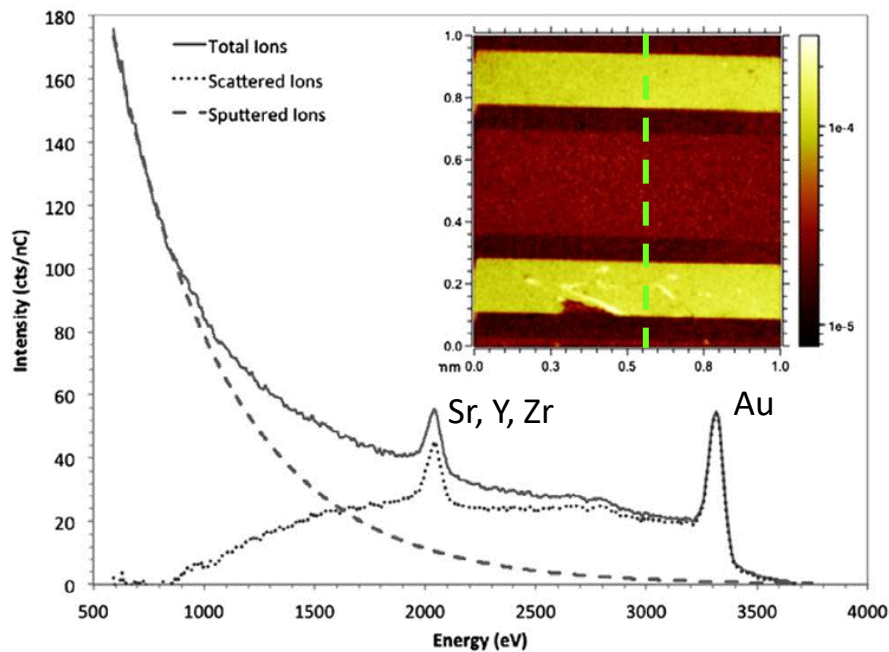


traditional porous electrode

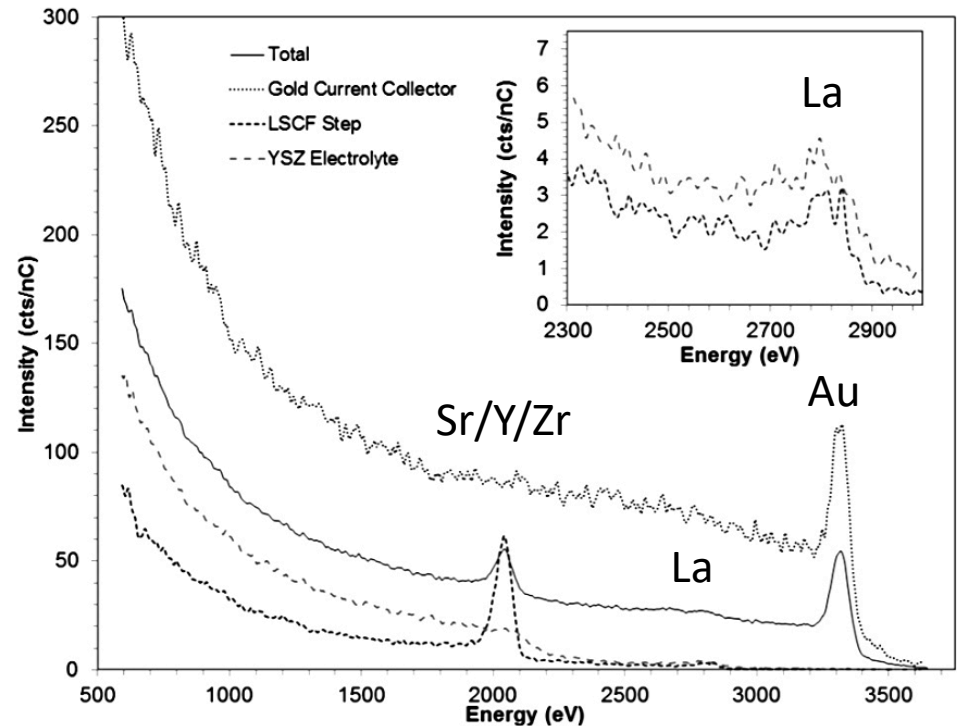
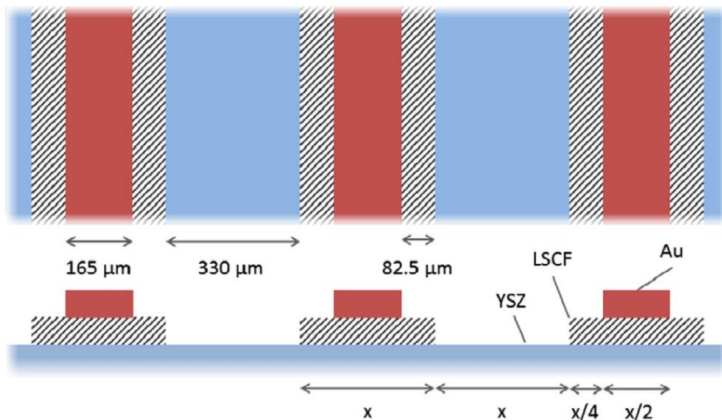


micropatterned LSCF electrodes on YSZ electrolyte

- > Laterally resolved analysis is possible (here: 5 keV Ne scattering)
- > Image resolution $\approx 5 \mu\text{m}$



- > LSCF step and electrolyte show no Au signal
- > LSCF mainly terminated by Sr
- > Electrolyte shows some La: Diffusion? Patterning? Electrochemical testing?
- > Electrolyte shows no Y or Zr
 - monolayer contamination by Na, Si, Ca segregated from bulk



- > Low energy ion scattering (LEIS) is the most surface sensitive technique available - top atomic layer characterisation
- > Static depth profiling provides detailed information up to 10 nm
- > LEIS provides straight-forward and matrix effect free quantification
- > The superior sensitivity of the Qtac100 double toroidal energy analyser allows real static LEIS analysis even with heavier projectiles at higher energies.
- > The time-of-flight mass filtering significantly improves detection limits
- > Ideal in combination with other analytical techniques such as TOF-SIMS or XPS



IONTOF

qtac¹⁰⁰

